

Improved performance of electric double layer capacitor using redox additive ($\text{VO}^{2+}/\text{VO}_2^+$) aqueous electrolyte†Cite this: *J. Mater. Chem. A*, 2013, **1**, 7913S. T. Senthilkumar,^a R. Kalai Selvan,^{*a} N. Ponpandian,^b J. S. Melo^c and Y. S. Lee^d

Electric double layer capacitors (EDLCs) were fabricated using biomass derived porous activated carbon as electrode material with 1 M H_2SO_4 and VOSO_4 added 1 M H_2SO_4 as electrolytes. Here, VOSO_4 was used as redox additive to improve the overall performance of EDLC. As expected, the VOSO_4 electrolyte showed ~43% of improved specific capacitance of 630.6 F g^{-1} at 1 mA cm^{-2} compared to pristine 1 M H_2SO_4 (440.6 F g^{-1}) due to the contribution of $\text{VO}^{2+}/\text{VO}_2^+$ redox reaction at the electrode–electrolyte interface. Possible redox reaction mechanism of $\text{VO}^{2+}/\text{VO}_2^+$ pair is also briefly illustrated. The good cycling performance of 97.57% capacitance retention was observed even after 4000 cycles. For comparison, the polymer gel electrolyte ($\text{PVA}/\text{VOSO}_4/\text{H}_2\text{SO}_4$) was also prepared and then the performance of the fabricated EDLCs was studied. Overall, these findings could open up a simple and cost effective way to improve the performance of EDLCs significantly.

Received 9th March 2013

Accepted 22nd April 2013

DOI: 10.1039/c3ta10998d

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1 Introduction

In recent times the electric double layer capacitors (EDLCs) have been identified as one of the foremost energy storage devices due to their versatile properties like good safety, fast charging–discharging, long cycle life and higher power density than batteries.^{1,2} However, the EDLCs (*i.e.*, carbon based materials as electrode materials) have delivered the nominal specific capacitance ($\sim 150 \text{ F g}^{-1}$) and energy density ($4\text{--}5 \text{ W h kg}^{-1}$) in liquid electrolytes (*i.e.*, aqueous and alkali) due to their limiting factor of low water decomposition potential of 1.23 V.³ In order to enhance the performance of EDLCs generally two different methods have been adopted in the carbon based materials. In one, heteroatoms such as oxygen,⁴ nitrogen,⁴ phosphorous⁵ and boron⁶ are introduced into the carbon materials and in the second, composites are made with carbon materials like carbon-metal oxide ($\text{Ni}(\text{OH})_2/\text{MWCNT}$, MgO/CNT and $\text{MnO}_2/\text{graphene}$)^{7a-c} and carbon-conducting polymer ($\text{PANI}/\text{carbon sphere}$, $\text{PANI}/\text{graphene}$, $\text{PANI}/\text{carbon nanocoil}$, $\text{PPY}/\text{carbon nanocoil}$ and PPY/GO).^{8a-d} However, both the methods have their own disadvantages including declining carbon conductivity, poor cycle

stability and multi-step process of composite preparation. In order to overcome these difficulties, very recently an innovative approach of adding redox species into the electrolytes (*i.e.*, redox electrolyte) was introduced to improve the performance of EDLCs *via* electron transfer at electrode–electrolyte interface through reversible redox reactions. This technique is very simple and cost effective approach compared with composite preparation. Organic compounds are mostly used as redox-active compounds due to their better solubility on electrode surface during redox process. Notably, the hydroquinone redox-active electrolyte delivered the maximum specific capacitance of 901 F g^{-1} unfortunately this system retained 65% of its initial specific capacitance after 4000 cycles. Similarly, indigo carmine,² methylene blue⁹ and sodium lignosulfonate,¹⁰ *p*-phenylenediamine^{11,12} and *m*-phenylenediamine¹³ have also been reported as redox-active compounds but these have not reached the specific capacitance of hydroquinone redox-active electrolyte (901 F g^{-1}). Thus the impetus is to find new or alternative redox additives to improve the specific capacitance with better cycle performance.

Here we present an alternative approach of using inorganic compound, VOSO_4 as a source of VO^{2+} *i.e.*, redox additive in H_2SO_4 electrolyte. It is well known that $\text{VO}^{2+}/\text{VO}_2^+$ is one of the most important ionic species in vanadium redox flow batteries because it exhibits the reversible redox reaction at electrode–electrolyte interface and improves their energy storage performance.^{14a} Also, their abundance, economical and excellent solubility in aqueous medium favors its use. But, the $\text{VO}^{2+}/\text{VO}_2^+$ redox reaction has slower kinetics compared to $\text{V}^{3+}/\text{V}^{2+}$.^{14b} However, the redox reaction of $\text{VO}^{2+}/\text{VO}_2^+$ is enhanced through catalytic process using hydroxyl or carbonyl functionalized carbon materials^{14a,c} and carbon supported CuPt_3 (ref. 14b) as electrodes resulting in a better improved $\text{VO}^{2+}/\text{VO}_2^+$ redox reaction in

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† Electronic supplementary information (ESI) available: See DOI: 10.1039/c3ta10998d

vanadium redox flow batteries. In this connection, here we tried to enhance the performance of the bio-waste derived activated carbon used EDLC using VOSO₄ added electrolyte through their VO²⁺/VO₂⁺ redox reaction. Also, here it is found that the VOSO₄ added 1 M H₂SO₄ electrolyte showed ~43% of improved specific capacitance of 630.6 F g⁻¹ at 1 mA cm⁻² compared to pristine 1 M H₂SO₄ (440.6 F g⁻¹) and good cycling performance (97.57% retention) was also obtained for 4000 cycles.

2 Experimental section

The activated carbons (ACs) were prepared from *Eichhornia crassipes* by chemical activation using NaOH as reported by us earlier.¹⁵ Briefly, the dried pieces of *Eichhornia crassipes* were pulverized. Then, 20 g of pre-heated (at 200 °C for overnight) pulverized sample was activated in 10% of NaOH for 24 h. Subsequently, it was carbonized at different temperatures such as 600 °C (NaA-1), 700 °C (NaA-2) and 800 °C (NaA-3) for 2 h under Ar atmosphere. The carbonized samples were washed several times with distilled H₂O with desired amount of HCl until the pH reached neutral. The washed samples were dried at 100 °C for overnight.

The prepared ACs were studied by powder XRD using X'Pert PRO PANalytical X-ray Diffractometer. The surface area, pore size and pore distribution of the ACs were carried out using nitrogen adsorption-desorption experiments at 77 K (Micromeritics ASAP 2010 surface area analyser) and calculated by BET method and BJH method, respectively. FTIR spectra of the ACs were recorded with model, Bruker tensor 27. The X-ray photoelectron spectroscopy (XPS) was also examined for AC using a Perkin-Elmer PHI-5702. The microscopic porous structure of the AC was observed *via* scanning electron microscopy (SEM) (Quanta 200 ESEM, FEI, USA).

The electrodes for EDLCs were prepared by drop-cast method.^{16a} Initially, the slurry was prepared by mixing of AC (20 mg), carbon black (2 mg) and polyvinylidene fluoride (PVDF, 2 mg) in 0.4 mL of *N*-methyl-2-pyrrolidone (NMP). Further, 12 µL of this slurry was spread onto the stainless steel with an area of 1 cm² and dried at 50 °C for overnight. The active material loading was calculated to be 0.6 mg on each electrode (excluding carbon black and PVDF mass). Finally, EDLCs (1 M H₂SO₄ and VOSO₄ added 1 M H₂SO₄) were fabricated by electrode sandwich type where electrolyte (1 M H₂SO₄ or 0.3 g VOSO₄ added 1 M H₂SO₄) immersed polypropylene sheet was used as separator, individually.

To optimize the electrode, electrochemical properties of the AC electrodes were studied by cyclic voltammetry (CV) and galvanostatic charge-discharge (GCD) in 1 M H₂SO₄ using three electrode systems (AC-working electrode, Pt wire-counter electrode and Ag/AgCl-reference electrode). Subsequently, fabricated EDLCs were examined through CV, GCD, electrochemical impedance spectroscopy (EIS) and cycle life. Here, CV and GCD tests for EDLCs were performed in the potential range of 0 to 0.8 V at various scan rates (5–200 mV s⁻¹) and current densities from 1 to 20 mA cm⁻², respectively. Electrochemical impedance spectroscopy (EIS) measurements were accomplished by applying an AC voltage with 10 mV amplitude in the frequency

range from 100 mHz to 1 MHz at open circuit voltage (OCV). All the electrochemical studies were carried out with Bio-Logic SP150 at room temperature. Besides, the EC-Lab V10.12 software was used for fitting the EIS data.

The specific capacitance of cell (C_{cell}) and specific capacitance of single electrode (C_{SP}) was calculated using the following eqn,^{17a}

$$C_{\text{cell}} = \frac{I\Delta t}{M\Delta V} \text{ and } C_{\text{SP}} = 4C_{\text{cell}} \quad (1)$$

where, I is the applied current, M is the total mass of the active material (AC) in both electrodes, Δt is the discharge time, ΔV is the potential range after IR drop.

Similarly, the energy and power density and coulombic efficiency was calculated using the following eqn,^{17a,16b}

$$E_{\text{cell}} = \frac{C_{\text{cell}}\Delta V^2}{7.2}; P_{\text{cell}} = \frac{E_{\text{cell}}}{\Delta t} \text{ and } \eta = \frac{\Delta t_{\text{d}}}{\Delta t_{\text{c}}} \quad (2)$$

where, Δt_{d} is the discharging time and Δt_{c} is the charging time.

The average discharge current (I_{d}) from cyclic voltammogram, areal capacitance (C_{a}) and mean areal capacitance (C_{m}) were obtained using the following eqn,^{17a-c}

$$I_{\text{d}} = \frac{\int Idv}{2\Delta V}; C_{\text{a}} = \frac{I_{\text{d}}}{sA} \text{ and } C_{\text{m}} = \frac{\Delta I}{2\Delta s} \quad (3)$$

where, $\int Idv$ is the integral area of CV curve, ΔV is the voltage difference, s is the scan rate, A is the area of active material in electrolyte and $\frac{\Delta I}{\Delta s}$ is the slope value.

3 Results and discussion

Fig. 1 shows, the nitrogen (N₂) adsorption-desorption isotherm, pore size distribution and the relation between the pore size

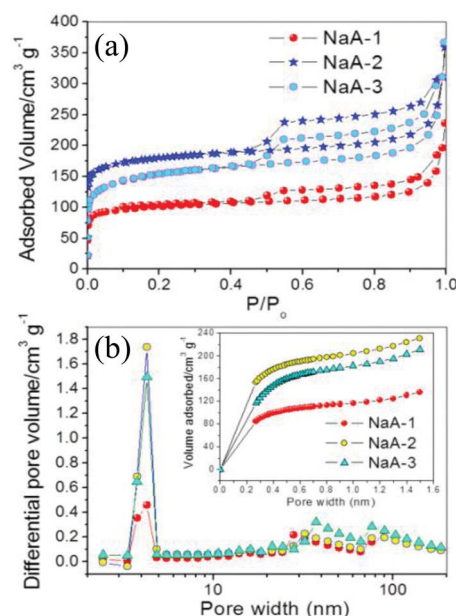


Fig. 1 (a) N₂ adsorption-desorption isotherm; (b) pore size distribution and pore width vs. volume adsorbed (inset) of NaA-1, NaA-2 and NaA-3.

and adsorbed pore volume of the prepared ACs at different temperatures such as 600 °C (NaA-1), 700 °C (NaA-2) and 800 °C (NaA-3). All the ACs show (Fig. 1a) the hysteresis loop at above $P/P_0 = 0.4$ between the adsorption-desorption branch which describes that the prepared ACs are mesoporous (2–50 nm) materials. The observed small tail at above $P/P_0 = 0.9$ indicates the presence of macropores (>50 nm). Also, at below $P/P_0 = 0.4$ is because of existence of micropores.¹⁹ The peaks centered at 4.3 nm, ~33 nm and ~82 nm (Fig. 1b) infer the presence of a combination of mesopores and macropores. Among the ACs, NaA-2 exhibits the highest BET surface area of 683.26 m² g⁻¹ compared to NaA-1 (381.64 m² g⁻¹) and NaA-3 (559.11 m² g⁻¹). Moreover among the ACs, NaA-2 shows a larger saturated adsorbed volume (Fig. 1b (inset)) which indicates a higher surface area.^{18–20}

The representative SEM, XRD and XPS images of NaA-2 are shown in Fig. 2. Fig. 2a illustrates the SEM image of AC which indicates the porous morphology and it is an evidence for the fact that AC has an interconnected porous structure. From XRD pattern (Fig. 2b), a strong carbon peak is observed in between 20° and 30° which corresponds to the (002) plane and indicates that the AC contains small disordered structure of aromatic sheets or disordered carbon. The other broad peak around 43° is attributed to the plane of (001) diffraction and indicates the presence of sp² hybridized carbon of aromatic structure.²¹ The FT-IR spectrum (Fig. S1†) shows the characteristic peak at ~3400 cm⁻¹ which corresponds to O–H stretching vibrations of hydroxyl group in phenol or carboxylic acid or adsorbed water. The broad peak at 2900–2100 cm⁻¹ is due to –C–H stretching. Similarly, the band at 1300–1030 cm⁻¹ is ascribed to –C–O stretching vibrations of hydroxyl or ether and O–H bending vibrations. The observed peaks at 1563 cm⁻¹ are attributed to stretching vibrations of carbonyl groups (C=O).^{22,23} In addition, the presence of surface functional groups was examined by XPS

and its representative C 1s and O 1s core level spectra of NaA-2 are shown in Fig. 2c and d. The obtained convoluted three peaks at 284.4 eV, 285.4 eV and 287.3 eV correspond to C 1s spectra and is attributed to presence of graphitic (C–C) or C–H groups, hydroxyl groups C–OH or and carbonyl/quinone groups (C=O). The O 1s core level spectra contains the four peaks such as 531.6 eV, 533.2 eV, 534.6 eV and 535.8 eV ascribed to the carbonyl/quinone groups (C=O), hydroxyl groups (C–OH) in phenol, hydroxyl groups in carboxylic (O=C–OH) acid and chemisorbed water, respectively.^{22,23} From the spectral analysis it can be concluded that the hydroxyl functional groups are mostly presented on NaA-2 sample.

In order to optimize the electrode, the CV and GCD tests are carried out for all the samples using a three electrodes system. The obtained CV curves (5 mV s⁻¹) and charge-discharge curves (1 mA cm⁻²) of AC electrodes (NaA-1, NaA-2 and NaA-3) are shown in Fig. 3. The quasi-rectangular CV curves are obtained for all the ACs, which is the characteristic feature of pseudo-capacitance. This behaviour is due to the reversible redox reaction between the oxygen functionalities like carbonyl (C=O), hydroxyl (C–OH, O=C–O) and H⁺.^{24,25} The presence of these kind of functional groups are identified through FT-IR and XPS analysis. Nonetheless, the NaA-2 electrode covers a larger current area in CV than NaA-1 and NaA-3 which reveals a higher ionic storage capacity which in turn reflects a higher capacitance. Similarly, the maximum specific capacitance of 587 F g⁻¹ was calculated from charge-discharge curve at 1 mA cm⁻² for NaA-2 while it is 528 F g⁻¹ and 420 F g⁻¹ for NaA-1 and NaA-3 respectively. This higher specific capacitance may be due to the high surface area of NaA-2 (683.26 m² g⁻¹). Overall, NaA-2 is deemed a better and optimal AC electrode for EDLC fabrication.

In order to identify the effect of VOSO₄ concentration in 1 M H₂SO₄, the cyclic voltammogram (CV) and electrochemical impedance spectroscopy (EIS) were carried out. The CV curves of NaA-2 electrode in different amounts of VOSO₄ (0.2 g, 0.3 g, 0.4 g and 0.5 g) added 1 M H₂SO₄ electrolyte is shown in Fig. 4a, which shows the quasi-rectangular behaviour. Fig. 4b shows the calculated discharge current behaviour of various content of VOSO₄ at different scan rates. The calculated areal capacitance are 134 mF cm⁻², 188 mF cm⁻², 164 mF cm⁻² and 172 mF cm⁻² at lower scan rate of 5 mV s⁻¹ and 384 mF cm⁻², 458 mF cm⁻², 369 mF cm⁻² and 377 mF cm⁻² at higher scan rate of 50 mV s⁻¹ for 0.2 g, 0.3 g, 0.4 g and 0.5 g of VOSO₄ respectively. However, the maximum areal capacitance at both lower and higher scan

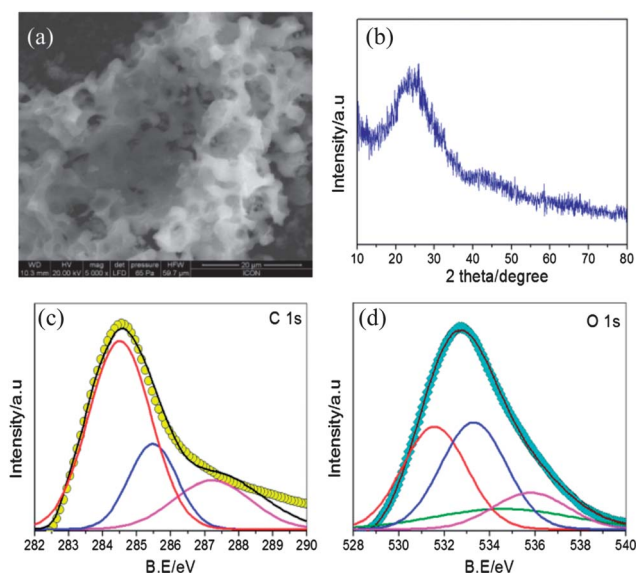


Fig. 2 The typical (a) SEM image; (b) XRD pattern; (c and d) C 1s and O 1s core level spectra of NaA-2.

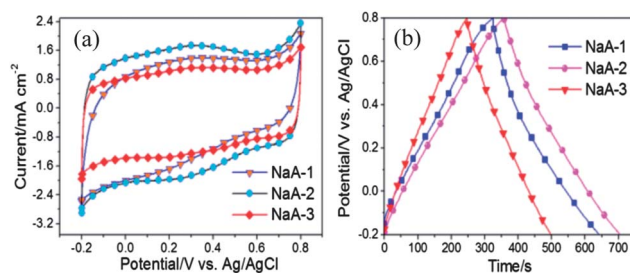


Fig. 3 Electrochemical characterization of AC (NaA-1, NaA-2 and NaA-3) electrodes: (a) CV curves at 5 mV s⁻¹; (b) charge-discharge curves at 1 mA cm⁻².

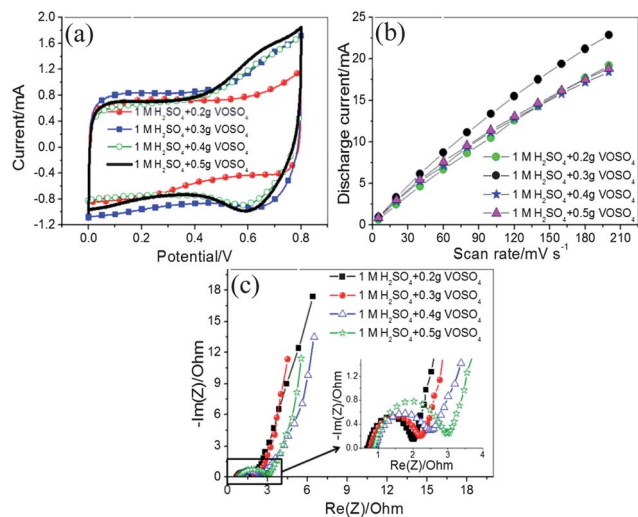


Fig. 4 (a) CV curves of NaA-2 electrode in different amount of VOSO_4 added 1 $\text{M H}_2\text{SO}_4$ electrolyte; (b) discharge current as function of scan rate; (c) Nyquist plot of VOSO_4 added 1 $\text{M H}_2\text{SO}_4$ electrolyte at different amount of VOSO_4 .

rate as well as better linear discharge current behavior was noted for 0.3 g of VOSO_4 , which indicates its better electrochemical properties. Moreover, the observed semicircle in Nyquist plot (Fig. 4c) is shifted along the impedance real axis as well as diameter of the semicircle is increased with increasing amount of VOSO_4 . This reveals the increased charge transfer resistance. However, due to the better discharge current behavior of 0.3 g VOSO_4 added H_2SO_4 electrolyte, it is considered as optimal concentration for EDLC and further studied as follows.

Fig. 5a shows the cyclic voltammogram of EDLCs using 1 $\text{M H}_2\text{SO}_4$, 1 $\text{M H}_2\text{SO}_4 + 0.3 \text{ g VOSO}_4$ and 1 M VOSO_4 as electrolyte at 5 mV s^{-1} . The rectangular-like shape CV curve observed for 1 $\text{M H}_2\text{SO}_4$ electrolyte, infers electric double layer capacitance and its interfacial capacitance was 90 mF cm^{-2} . Interestingly quasi-rectangular shaped CV curve with increased interfacial capacitance of 150 mF cm^{-2} was observed for VOSO_4 (0.3 g) added electrolyte. This quasi-rectangular behaviour is due to the redox reaction of $\text{VO}^{2+}/\text{VO}_2^+$. This was further confirmed through the CV (Fig. 5b) of EDLC using 1 M VOSO_4 as electrolyte, where it clearly indicates the well defined redox peak. Fig. 5c shows the discharge current response of EDLC using 1 $\text{M H}_2\text{SO}_4$ and VOSO_4 added 1 $\text{M H}_2\text{SO}_4$ as electrolyte with respect to scan rate. As expected higher discharge current was observed for the VOSO_4 with EDLC, which is due to the occurrence of redox reaction between $\text{VO}^{2+}/\text{VO}_2^+$. From Fig. 5c, the maximum mean areal capacitance of 55.8 mF cm^{-2} was calculated for VOSO_4 added 1 $\text{M H}_2\text{SO}_4$ electrolyte but it is only 37 mF cm^{-2} for 1 $\text{M H}_2\text{SO}_4$ electrolyte. So the combination of VOSO_4 (0.3 g) and 1 $\text{M H}_2\text{SO}_4$ could improve the performance of pristine electrolyte (1 $\text{M H}_2\text{SO}_4$) used EDLC through the possible redox reaction of $\text{VO}^{2+}/\text{VO}_2^+$ at electrode-electrolyte interface. The possible redox reaction is given below.

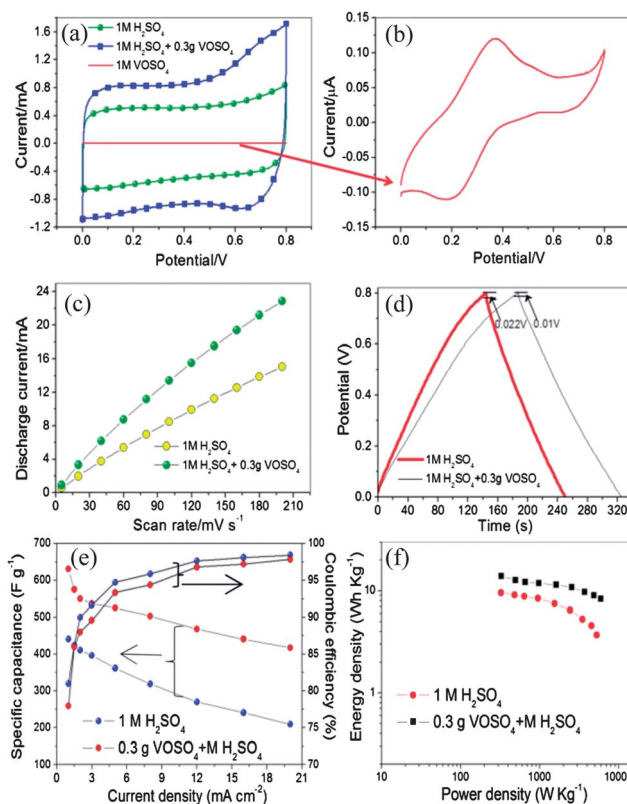
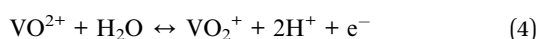
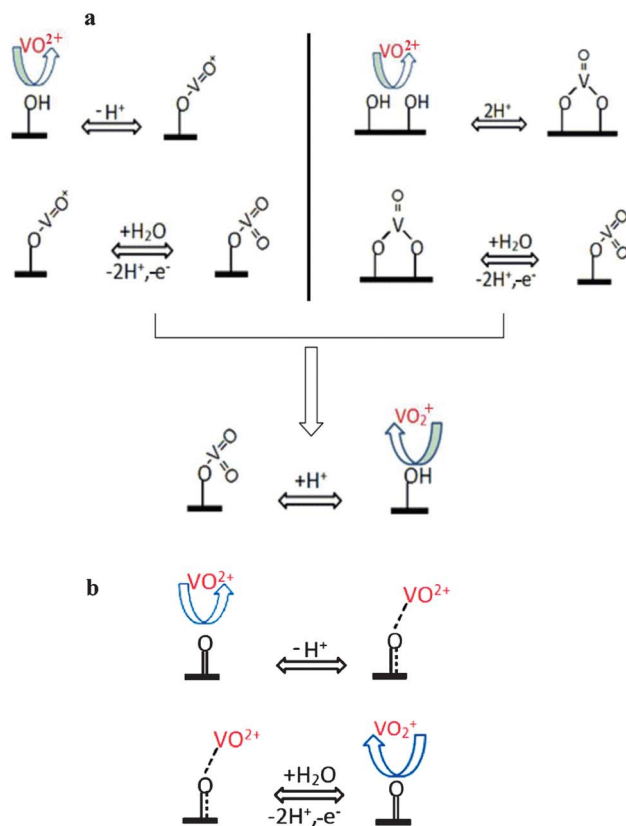


Fig. 5 Electrochemical characteristics of liquid electrolyte using EDLC: (a) comparison of CV curves of 1 M VOSO_4 , 1 $\text{M H}_2\text{SO}_4$ and $\text{VOSO}_4 + 1 \text{ M H}_2\text{SO}_4$ electrolyte at 5 mV s^{-1} ; (b) CV curve of 1 M VOSO_4 electrolyte for clarity; (c) scan rate as a function of discharge current; (d) comparison of charge-discharge curves of 1 $\text{M H}_2\text{SO}_4$ and $\text{VOSO}_4 + 1 \text{ M H}_2\text{SO}_4$ electrolyte at 1 mA cm^{-2} ; (e) specific capacitance and coulombic efficiency as a function of current rate and (f) Ragone plot.

Furthermore, the redox reaction of $\text{VO}^{2+}/\text{VO}_2^+$ is enhanced by catalytic reaction in presence of hydroxyl ($-\text{OH}$) and carbonyl ($\text{C}=\text{O}$) functional groups *via* oxygen transfer (Scheme 1). More specifically, during charging in the case of hydroxyl groups (Scheme a), the adsorbed VO^{2+} on the electrode replaces the H^+ from hydroxyl group and creates the bond. Then, the oxygen atom is transferred and forms VO_2^+ on the surface of electrode by releasing of H^+ and e^- . Finally, the VO_2^+ diffuses into the bulk solution *via* exchange of H^+ from bulk solution. In contrast, in the case of carbonyl groups (Scheme b) the adsorbed VO^{2+} creates the transition state with carbonyl groups. Subsequently, it reacts with H_2O while the VO_2^+ diffuses back into the bulk electrolyte as VO_2^+ through release of H^+ and e^- .^{26,27a} In addition, during discharge these reactions are reverse of the charging process. So, the hydroxyl and carbonyl functional groups also help the redox reaction of $\text{VO}^{2+}/\text{VO}_2^+$ *via* catalytic process.

The galvanostatic charge-discharge curves of EDLCs at 1 mA cm^{-2} are shown in Fig. 5d. The calculated specific capacitances from GCD are 440.6 and 630.6 F g^{-1} . Here, the 1 $\text{M H}_2\text{SO}_4 + \text{VOSO}_4$ electrolyte exhibits the higher specific capacitance with $\sim 43\%$ of improved specific capacitance than pristine 1 $\text{M H}_2\text{SO}_4$



Scheme 1 The redox reaction of $\text{VO}_2^+/\text{VO}_2^+$ during charging–discharging via catalytic activity with hydroxyl ($-\text{OH}$) (scheme a) and carbonyl ($\text{C}=\text{O}$) (scheme b) functional groups.

electrolyte. This enhanced specific capacitance further confirms the occurrence of redox reaction of $\text{VO}_2^+/\text{VO}_2^+$ at electrode–electrolyte interface. Moreover, this is higher than the specific capacitance of composite electrodes using symmetric supercapacitors for example PANI/GO (555 F g^{-1})^{27b} as well as some redox additive electrolyte using EDLCs such as *m*-methylene blue (279 F g^{-1}),⁹ liginosulfonates (178 F g^{-1}),¹⁰ *m*-phenylenediamine (78 F g^{-1})¹⁴ and *p*-phenylenediamine (605.33 F g^{-1}).¹²

Fig. 5e shows that the specific capacitance decreases with increasing current density. However, relatively 66.2% of capacitance (417.5 F g^{-1}) retention was found for VOSO_4 added 1 M H_2SO_4 electrolyte at higher current density (20 mA cm^{-2}) while it was about 47.5% (209.2 F g^{-1}) for pristine 1 M H_2SO_4 . Besides, the VOSO_4 added 1 M H_2SO_4 electrolyte shows the low coulombic efficiency (77.3%) compared to pristine electrolyte (80.9%) at 1 mA cm^{-2} whereas this result could be associated with kinetics of $\text{VO}_2^+/\text{VO}_2^+$ redox reaction. However, both electrolytes show the maximum coulombic efficiency of $\sim 98\%$ at higher current density (97.8% for VOSO_4 added 1 M H_2SO_4 electrolyte and 98.4% for pristine electrolyte at 20 mA cm^{-1}). Interestingly, the IR drop decreased from 0.022 V to 0.01 V (Fig. 5d) due to the effect of VOSO_4 in 1 M H_2SO_4 . This decreasing IR drop value can also improve the performance of the EDLC, based on the relation^{28,29a} $E = 0.5C [(V_{\text{initial}} - \text{IR})^2 - V_{\text{final}}^2]$. If IR drop is high, energy loss is unavoidable. The

equivalent series resistance (ESR) of the EDLCs are also calculated¹⁴ ($\text{ESR} = \text{IR}_{\text{drop}}/2I$, where I is current density) from charge–discharge curve and obtained the lower ESR value of 5Ω over pristine 1 M H_2SO_4 (11Ω) and it indicates the improved ionic conductivity in electrolyte.

Nevertheless, a higher energy density of 13.7 W h kg^{-1} was obtained than with pristine electrolyte (9.3 W h kg^{-1}) at 1 mA cm^{-2} . It is nearly 47% of energy density increment than pristine (Fig. 5f). This obtained energy density is even comparable and higher than that of previous redox-active electrolyte using EDLCs such as indigo carmine (1.7 W h kg^{-1}),² *m*-phenylenediamine (9.99 W h kg^{-1})¹⁴ and *p*-phenylenediamine (4.23 W h kg^{-1}).¹³ As well as higher than the reported asymmetric supercapacitors such as $\text{TiP}_2\text{O}_7//\text{AC}$ (13 W h kg^{-1}),^{29b} $\text{TiO}_2 (\text{B})//\text{CNT}$ (12.5 W h kg^{-1})^{29c} PEDOT// MnO_2 (13.5 W h kg^{-1}),³² AC/MnO_2 (10.4 W h kg^{-1}),³³ $\text{CNT}/\text{graphene}/\text{MnO}_2$ (12.5 W h kg^{-1}),³⁴ $\text{AC}/\text{RuO}_2/\text{TiO}_2$ (5.7 W h kg^{-1}),³⁵ $\text{MnO}_2/\text{Fe}_3\text{O}_4$ (8.1 W h kg^{-1})³⁶ and PPy/PMA//PEDOT/PTA (4 W h kg^{-1}).³⁷ Unfortunately, there is no significant increment in power density after introducing VOSO_4 *i.e.*, 315.2 W Kg^{-1} for 1 M H_2SO_4 electrolyte and 325 W Kg^{-1} for $\text{VOSO}_4 + 1 \text{ M H}_2\text{SO}_4$ electrolyte. This may due to the decreasing IR drop or ESR after addition of VOSO_4 .

The Nyquist impedance and admittance spectra of EDLCs are shown in Fig. 6a and Fig. S4a†, respectively. The solution resistance (R_s) and charge transfer resistance (R_{ct}) of EDLCs was calculated by fitting of EIS using equivalent circuit and is shown in inset of Fig. 6a, which includes a solution resistance (R_s), charge transfer resistance (R_{ct}), electrical double layer element (C_{edl}), pseudocapacitance element (C_{pse}) and Warburg impedance (W). Here, electrical double layer element (C_{edl}), pseudocapacitance element (C_{pse}) indicates the presence of capacitance and pseudocapacitance through adsorption of electrolyte's ions and electron-transfer or redox reaction of electrolyte's ions. The calculated solution resistance of 1 M H_2SO_4 and $0.3 \text{ g VOSO}_4 + 1 \text{ M H}_2\text{SO}_4$ electrolytes are 1.5Ω and 0.8Ω whereas the measured charge transfer resistance in the electrodes are 3Ω and 1.6Ω respectively. However, the addition of VOSO_4 in 1 M H_2SO_4 reduces the charge transfer resistance and reveals the improved ionic conductivity at electrode–electrolyte interface. Moreover, the small diameter semicircle is noted for $\text{VOSO}_4 + 1 \text{ M H}_2\text{SO}_4$ electrolyte which characterizes its lower ionic resistance.³⁰ Similarly, the knee frequency also increased (Fig. S4a†) from 25.36 Hz to 53.98 Hz for the VOSO_4

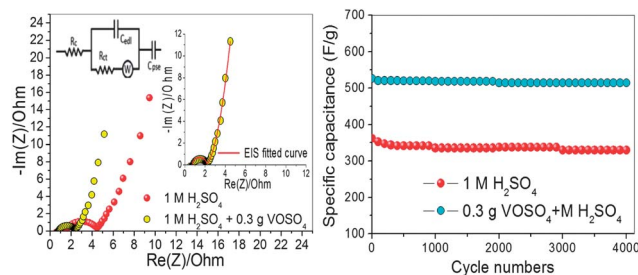


Fig. 6 (a) Nyquist plots (inset: equivalent circuit and fitted EIS curve); (b) variation of specific capacitance with respect to cycles at 5 mA cm^{-2} of 1 M H_2SO_4 and $\text{VOSO}_4 + 1 \text{ M H}_2\text{SO}_4$ electrolyte.

added electrolyte which also denoted the improved performance.³¹ Fig. S4b† shows the real part capacitance (C') as a function of frequency of EDLCs. As well, it describes the increased capacitance after adding VOSO₄ in 1 M H₂SO₄ electrolyte at low frequency which is consistent with the results of CV and GCD. The variation of specific capacitance with cycle numbers at 5 mA cm⁻² is shown in Fig. 6b. The specific capacitance of VOSO₄ + 1 M H₂SO₄ electrolyte decreases from 527 to 514.2 F g⁻¹ with better capacitance retention of 97.57% than 1 M H₂SO₄ electrolyte (91%) after 4000 cycles.

It is well known that liquid electrolytes have issues in packing, leakage, corrosion and so on. In this regard, currently the gel electrolytes are more interesting for EDLCs applications. Here, VOSO₄ + 1 M H₂SO₄ electrolyte was used to prepare the gel electrolyte. Subsequently, the EDLC was fabricated using VOSO₄ + 1 M H₂SO₄ added gel polymer electrolyte (for details see ESI†). Fig. 7a–d shows the CV at 5–200 mV s⁻¹, GCD at 1–20 mA cm⁻², specific capacitance as function of current density and output performance of the polymer gel electrolyte used in the EDLC. The CV retains its shape even at high scan rate of 200 mV s⁻¹, which reveals a better electrochemical retention property. From GCD, the maximum specific capacitance of 585 F g⁻¹ with energy density of 12.83 W h kg⁻¹ is achieved at 1 mA cm⁻². Here the decreased specific capacitance or energy density may be due to the lack of better contact between the porous electrode surface and polymer electrolyte. However, it is a very common problem for polymer electrolyte used in EDLCs and also these obtained values are comparable with the liquid electrolyte. In addition, this EDLC exhibits the coulombic efficiency of 63.8% and 97% at 1 mA cm⁻² and 20 mA cm⁻² (Fig. 7c), respectively. The charged EDLC (Fig. 7d) at 1 mA cm⁻² delivered the output voltage of 0.68 V (see video, ESI†). So this approach opens up a new avenue for the possible application especially for portable electronics.

4 Conclusions

Porous activated carbon was prepared from *Eichhornia crassipes* by NaOH activation. The EDLC using VOSO₄ added 1 M H₂SO₄ electrolyte showed maximum specific capacitance of 630.6 F g⁻¹ at 1 mA cm⁻² which is ~43% of improved specific capacitance than 1 M H₂SO₄ (440.6 F g⁻¹). Subsequently, the energy density also increased from 9.3 to 13.7 W h kg⁻¹ at 1 mA cm⁻². The IR drop or ESR value also decreased after adding VOSO₄ in 1 M H₂SO₄. From the above results it is confirmed that the redox pair VO²⁺/VO₂⁺ can enhance the overall electrochemical properties of EDLC. These results could establish that VOSO₄ added 1 M H₂SO₄ is an excellent candidate to improve the performance of the EDLCs.

Acknowledgement

The authors are grateful to the Department of Atomic Energy–Board of Research in Nuclear Sciences (DAE–BRNS), Government of India (no. 2010/37P/46/BRNS/1443) for their financial support.

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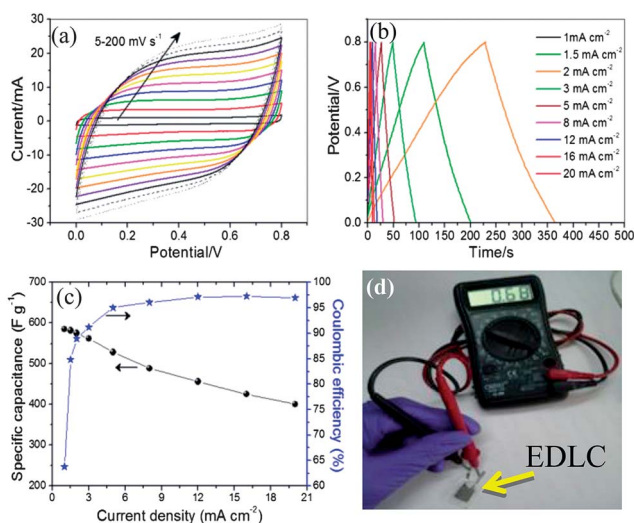


Fig. 7 Electrochemical characteristics of gel polymer electrolyte used in EDLC: (a) CV curves at 5–200 mV s⁻¹; (b) charge–discharge discharge curves at 1–20 mA cm⁻²; (c) specific capacitance and coulombic efficiency as function of current density; (d) performance of polymer electrolyte used EDLC.

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